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Baek, Su-Hyun

Department of Molecular and Material Sciences, Interdisciplinary Graduate School of
Engineering Sciences, Kyushu University

Inada, Miki

Department of Molecular and Material Sciences, Interdisciplinary Graduate School of
Engineering Sciences, Kyushu University

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Su-Hyun Baek¹ and Miki Inada^{1,2,3}

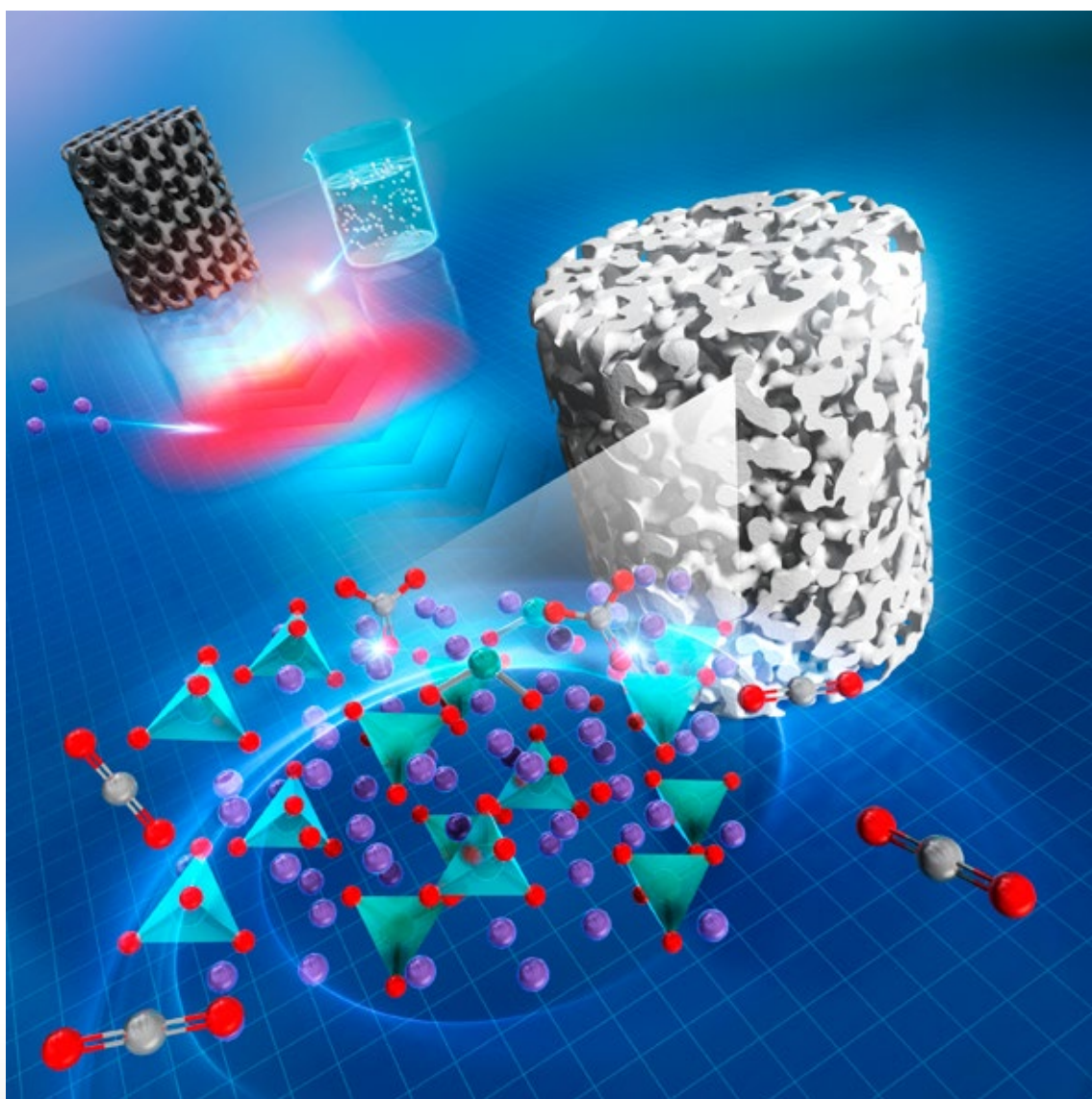
¹Department of Molecular and Material Sciences, Interdisciplinary Graduate School of Engineering Sciences, Kyushu University, 6-1 Kasuga-Koen, Kasuga, Fukuoka 816-8580, Japan

²Center of Advanced Instrumental Analysis, Kyushu University, 6-1 Kasuga-Koen, Kasuga, Fukuoka 816-8580, Japan

³Department of Applied Chemistry, Faculty of Engineering, Kyushu University, 744 Motoooka, Fukuoka 819-0395, Japan

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FULL PAPER

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Su-Hyun Baek¹ and Miki Inada^{1,2,3,†}

¹Department of Molecular and Material Sciences, Interdisciplinary Graduate School of Engineering Sciences, Kyushu University, 6–1 Kasuga-Koen, Kasuga, Fukuoka 816–8580, Japan

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³Department of Applied Chemistry, Faculty of Engineering, Kyushu University, 744 Motoooka, Fukuoka 819–0395, Japan

Lithium silicate (Li₄SiO₄) is one of the most promising carbon dioxide (CO₂) capture materials at high temperatures. In this study, Li₄SiO₄ on silica (SiO₂) porous structure was tried to fabricate in solid state reaction for CO₂ capture materials. Li₂CO₃ reacted with amorphous SiO₂ on the surface of silica porous structure made by template method, resulting in the formation of Li₄SiO₄ porous structure on silica support at 800 °C. The obtained Li₄SiO₄ porous structure can capture CO₂ at 600 °C and release it at 700 °C, and could be used stably and repeatedly. This Li₄SiO₄ porous structure is expected to be useful as a CO₂ capture material at high temperature.

Key-words : Lithium silicate, CO₂ sorbent, CO₂ capture, Surface modification

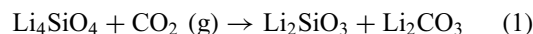
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1. Introduction

The amount of carbon dioxide (CO₂) emitted from coal-fired power plants is more than 20 % worldwide, and about 60 % in the power generation sector. Because coal has abundant reserves and is cheaper than oil or natural gas, its use is increasing.¹⁾ CO₂ is greenhouse and acidic gas causing to global warming and acidification of seawater.^{2)–4)} The reducing CO₂ emissions from coal-fired power plants is an important task.^{5),6)} One solution of these problems is a CO₂ removal method using a capture material.⁷⁾ Currently, various studies on CO₂ capture and storage (CCS) for reduction of CO₂ are being conducted around the world.^{8)–10)} One of the CCS technologies is using absorption and adsorption. The selection of capture material is important to improve CO₂ storage efficiency.¹¹⁾

The most practical method is chemical absorption reactions using amine solutions such as mono-ethanol amine, di-ethanol amine, and methyl diethylanamine, which has advantages for large-capacity processing and simple operation.¹²⁾ However, this technology requires a cooling system because high-temperature CO₂ gas cannot be taken directly from flue gas or exhaust gas, caused to an increase in energy consumption and requirement of time and money. In addition, there are problems with environmental hazards due to the volatility of the absorbent and the corrosiveness of the absorbent itself.¹³⁾ Therefore, it is necessary to develop a capture material at high temperatures. There are some literatures on calcium oxide (CaO) and lithium

silicate (Li₄SiO₄) as high-temperature solid capture materials.^{14)–16)} CaO has a high CO₂ capture capacity and wide availability, although its low cycle stability due to serious sintering problems.¹⁷⁾ Li₄SiO₄ has a high lithium atom density, excellent chemical and mechanical stabilities at high temperatures, and low neutron activation properties,^{18),19)} which has widely applied for catalysts, CO₂ capture materials, CO₂ separation membranes, cathodes in Li-ion batteries, and energy storage.^{20)–24)} Li₄SiO₄ has a high selective CO₂ capture ability with a theoretical value of 36.7 wt % CO₂ in the temperature range of 450–700 °C according to the reaction shown in Eq. (1).²⁵⁾ The regeneration temperature of Li₄SiO₄ is less than 750 °C, which is lower than that of CaO, the energy consumption required for regeneration is low, and circulation stability is excellent.^{26),27)}



There are some studies on kinetic analysis for the carbonation of Li₄SiO₄.^{28)–36)} Venegas et al. proposed two rate constants during CO₂ capture of Li₄SiO₄; one is a reaction rate constant between CO₂ and Li₄SiO₄, and the other is a diffusion rate constant of CO₂ into Li₂SiO₃ and Li₂CO₃ layer.²⁸⁾ This kinetic analysis can help us understand the CO₂ capture ability of Li₄SiO₄.

Regarding the reversible capture of CO₂, there are two main approaches with the use of liquid capture materials and solid capture materials.^{37),38)} The method of using liquid capture materials has a limitation due to the decomposition of liquid during temperature cycles. The solid capture materials are relatively non-degradable, have a variety of capture functions and are renewable.^{37)–39)} Most previously studied Li₄SiO₄ capture materials are not suitable

[†] Corresponding author: M. Inada; E-mail: inada.miki.300@m.kyushu-u.ac.jp

ble for application in various fields due to easy elution and rapid pressure drop when used in a reactor as powder form.^{40)–42)} Therefore, in order to be commercialized, development in bulk form is necessary. The capture materials of bulk form can be applied by strengthening their mechanical properties, which is a necessary element in renewable CO_2 capture materials.⁴³⁾ These bulk capture materials can be divided into pellet form, monolithic (honeycomb), and porous form. Monolithic and porous forms are more advantageous for separation and recovery, and have high mass transfer characteristics with low pressure drop.^{44)–49)} Therefore, the development of capture materials such as monolithic and porous materials has been active in recent years.^{46),47),50),51)} Inorganic porous capture materials have received industrial interest for use in CCS.^{52)–55)} The presence of capture sites for CO_2 is the most important factor. The form, particle size, porosity, and specific surface area of the material are also important to determine capture efficiency.^{56),57)} In particular, a large surface area is effective in higher capture amounts and longer capture life.⁵⁸⁾

In this study, we attempted to fabricate a renewable high temperature CO_2 capture material, Li_4SiO_4 , with a porous structure. There are few studies reporting the effectiveness and performance of CO_2 removal by Li_4SiO_4 capture material surface-modified on a porous support. Therefore, this study is meaningful in investigating the improvement effect of Li_4SiO_4 capture material surface-modified on SiO_2 porous sintered body. Li_4SiO_4 was synthesized on the surface of the SiO_2 porous glass support by an impregnation method. The characteristics and CO_2 capture ability of the Li_4SiO_4 porous sample were evaluated. The rate constant during the CO_2 capture process was obtained, and the effect of the porous structure on the reaction was investigated.

2. Experimental

2.1 Preparation of SiO_2 porous glass

The schematic diagram for sample preparation is shown in Fig. 1. SiO_2 ceramics of compositions 89 wt % SiO_2 (SS-10, Tokuyama Corp., Japan), 10 wt % mullite (KM101, KCM Corp., Japan), and 1 wt % K_2CO_3 (Special grade,

KISHIDA CHEMICAL Co., Ltd., Japan) were used as starting materials for the preparation of porous glass supports. SiO_2 , mullite and K_2CO_3 powders were mixed for 50 min using an agate mortar. A cubic open cell PU sponge (MF-30, INOAC Corp., Japan) with a cell density of 30 ppi was used as a template. Poly vinyl alcohol (PVA, special grade, FUJIFILM Wako Pure Chemical Corp., Japan) aqueous solution was used as a dispersant.

SiO_2 porous glass was prepared according to our previous report.⁵⁹⁾ 3 g of 5 wt % PVA aqueous solution was mixed with 3 g of SiO_2 powder using a kneading machine (AR-100, Thinky Corp., Japan). PU sponge with a diameter and height of 11.5 and 10.0 mm was dipped into the SiO_2 slurry before drying at room temperature for 24 h. The dried sample was calcined at 400 °C for 6 h, sintered at 1200 °C under atmospheric pressure with a heating rate of 10 °C/min. The obtained SiO_2 porous glass shrank to about 9.5 mm of diameter and 8.0 mm of height due to sintering.

2.2 Surface modification of SiO_2 porous glass with Li_4SiO_4

For the preparation of Li_4SiO_4 on SiO_2 porous glass named as $\text{Li}_4\text{SiO}_4/\text{SiO}_2$, Li_2CO_3 (Special grade, KISHIDA CHEMICAL Co., Ltd., Japan) was used as Li source. Li_2CO_3 slurry was prepared in the same way as SiO_2 slurry using 5 wt % PVA aqueous solution. The mixing composition was set at 6 g of 5 wt % PVA and 4 g of Li_2CO_3 powder. SiO_2 porous glass was dipped into the Li_2CO_3 slurry, and dried at room temperature for 24 h. The dried sample was heat-treated at 400 °C for 1 h and then at 800 °C for 1 h under atmospheric pressure at a heating rate of 10 °C/min.

2.3 Characterization

Phase analysis was performed using X-ray diffraction (XRD, SmartLab SE, Rigaku Corp., Japan). Amorphous and crystalline phase composition were calculated by Rietveld analysis. The microstructure of the sintered body was observed using JSM-6701F (JEOL Ltd., Japan).

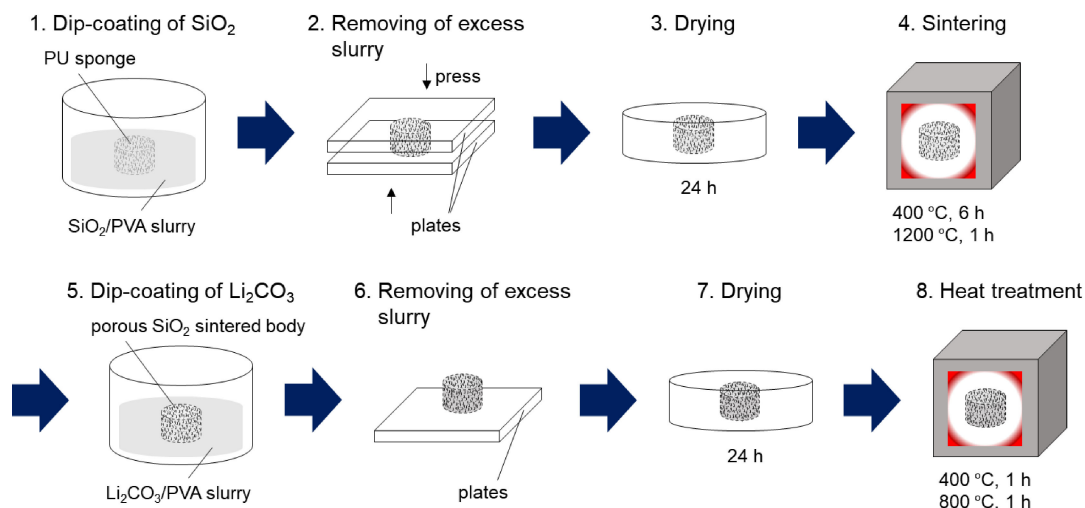


Fig. 1. Fabrication process of Li_4SiO_4 porous sorbent having a hierarchical structure using PU sponge.

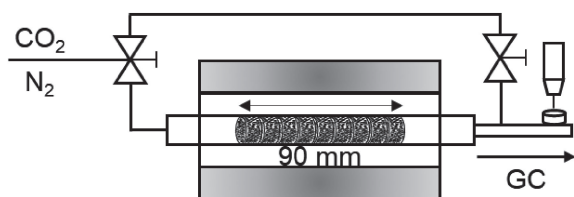


Fig. 2. Schematic diagram of the packed-bed system used for the CO₂ capture and regeneration experiments.

2.4 CO₂ capture and regeneration tests of Li₄SiO₄

CO₂ capture and regeneration tests were conducted in a fixed-bed stainless steel pipe reactor (internal diameter: 10 mm) heated by a tubular furnace. An overview of the device structure is shown in Fig. 2. The Li₄SiO₄/SiO₂ samples were filled in the stainless steel pipe reactor and heated at 600 and 700 °C during CO₂ capture and regeneration test. The flow gas condition was set with 0–15 vol % CO₂/N₂ balance at 100 or 200 mL/min under atmospheric pressure. The CO₂ concentration was analyzed at 5 min intervals by gas chromatography (GC-2014, Shimadzu Corp., Japan) with flame ionization detector (FID) mode using methanizer. The temperature was set at 100 °C as injector, 75 °C as Sunpak-A column, 150 °C as FID detector, and 150 °C as methanizer, respectively. The flow rate of N₂ as the carrier gas was set to 20.0 mL/min.

3. Results and discussion

3.1 Li₄SiO₄ surface modification conditions

The sintering of Li₄SiO₄ is quite difficult due to a low diffusion coefficient and thermal decomposition even at high temperature. In almost all cases, voids form and the densification do not occur.^{59),60)} In this study, we focused on the surface modification of Li₄SiO₄ to SiO₂ porous support. For the application to CO₂ capture materials, the surface activity is required compared to bulk property. Two methods were tried to modify the surface of SiO₂ porous support to Li₄SiO₄: one was coating with Li₄SiO₄ on SiO₂ support named as Li₄SiO₄ on SiO₂, and the other was the reaction of Li₂CO₃ with SiO₂ surface named as Li₄SiO₄/SiO₂. Li₄SiO₄ and Li₂CO₃ slurries were prepared by mixing of PVA with each powder. Li₄SiO₄ powder was made by solid state reaction.⁶¹⁾ The calcination at 400 °C for 1 h was carried out on both samples to remove PVA, and then the heat treatment was applied to Li₄SiO₄/SiO₂ sample at 800 °C for 1 h.

To confirm that Li₄SiO₄ was coated on the SiO₂ surface, the sample was cut and the SEM image was checked. Figure 3 shows the SEM images of cross-section view of (a) Li₄SiO₄ on SiO₂ and (b) Li₄SiO₄/SiO₂. The interface of Li₄SiO₄ and SiO₂ was clearly observed on coating sample, whereas that of reacting one was ambiguous. This suggests that Li₂CO₃ and SiO₂ were reacted very well, resulting in tight bonding. The high reactivity between Li₂CO₃ and SiO₂ might be due to the high activity of SiO₂ glass surface sintered at low temperature. Thus, silica porous materials with amorphous phase prepared by sintering is

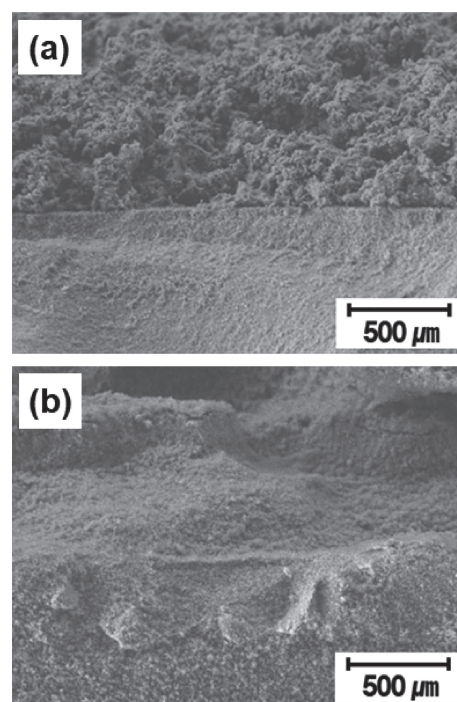


Fig. 3. SEM images of cross-section view of (a) Li₄SiO₄ on SiO₂ and (b) Li₄SiO₄/SiO₂.

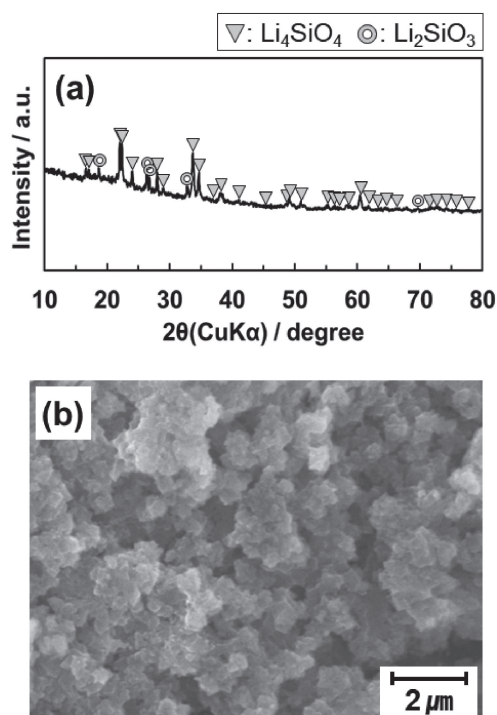


Fig. 4. (a) XRD pattern and (b) SEM image of Li₄SiO₄/SiO₂ surface.

effective for the fabrication of environmental catalyst by surface modification.

Figure 4 shows the XRD pattern and SEM image of Li₄SiO₄/SiO₂ surface prepared by heating at 800 °C for 1 h Li₂CO₃ on silica porous material. Li₄SiO₄ mainly formed beside small amount of Li₂SiO₃. The crystalline phase con-

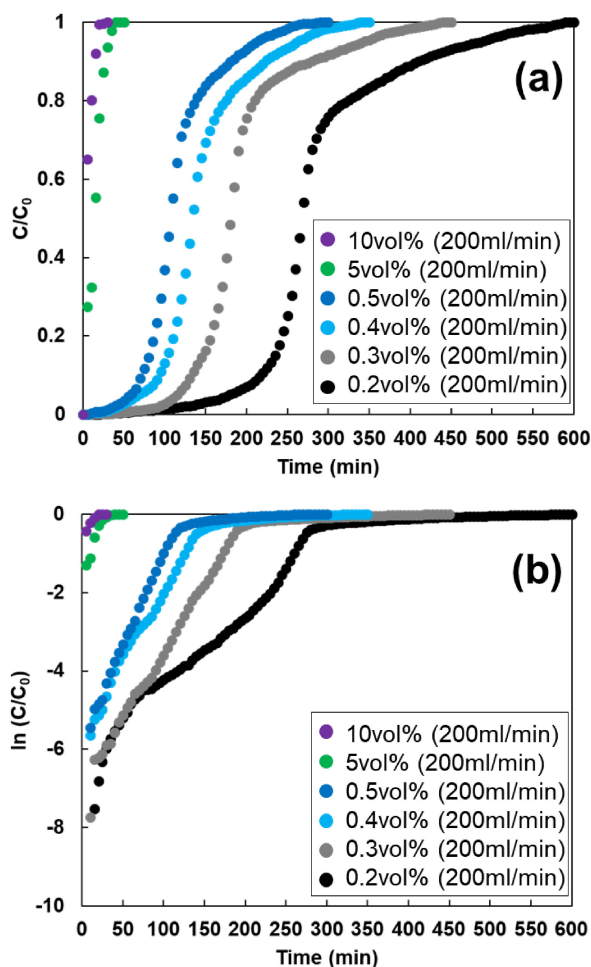


Fig. 5. CO_2 capture performance [C/C_0 -time (a) and $\ln(C/C_0)$ -time (b)] of prepared $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ porous samples with different initial concentration of CO_2 at 200 mL/min at 600 °C.

tent was 88 wt % (0.00063 mol) for Li_4SiO_4 and 12 wt % (0.00011 mol) for Li_2SiO_3 estimated using the Rietveld method. The particles of about 500 nm were observed on the surface of $\text{Li}_4\text{SiO}_4/\text{SiO}_2$, indicating that Li_4SiO_4 was synthesized by solid reaction of Li_2CO_3 and SiO_2 .

3.2 CO_2 capture characteristics of $\text{Li}_4\text{SiO}_4/\text{SiO}_2$

The CO_2 capture ability of the $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ porous sample was measured using 0.2–10 vol % CO_2 gas with N_2 balance of 200 mL/min at 600 °C. **Figure 5** shows the change in CO_2 concentration at reactor outlet with time, which means break through curve. The outlet CO_2 concentration C normalized by initial concentration C_0 . $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ porous sample can capture CO_2 until Li_4SiO_4 completely changed to Li_2CO_3 . When CO_2 initial concentration was high over 5 vol %, CO_2 was immediately permeated the test samples. Below 0.5 %, the increase in CO_2 concentration changed from gradually to sharply and gradually again with time, forming an S-curve. Especially at 0.2 and 0.3 %, CO_2 was completely captured during the initial 5–10 min of the reaction. These results indicate that CO_2 and Li_4SiO_4 react rapidly at the initial stage of reac-

Table 1. Rate constant values calculated from the $\ln(C/C_0)$ -time slope

Flow rate (mL/min)	Concentration (vol %)	k_1 (s^{-1})	k_2 (s^{-1})
200	0.2	0.0075	0.0058
	0.3	0.0075	0.0057
	0.4	0.0075	0.0058
	0.5	0.0075	0.0058
	5	—	0.0058
	10	—	0.0058

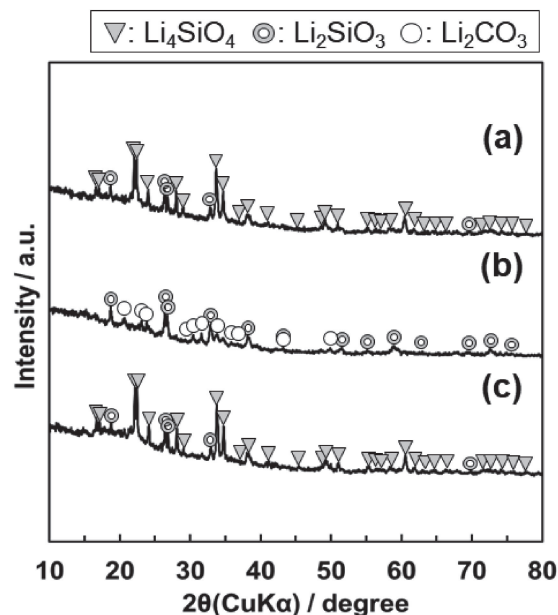


Fig. 6. XRD patterns of (a) $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ porous sample after (b) CO_2 capture at 600 °C and (c) regeneration at 700 °C.

tion and the Li_2CO_3 formed on the surface of Li_4SiO_4 prevent the reaction of CO_2 and Li_4SiO_4 from middle to final stage of reaction. CO_2 should diffuse the Li_2CO_3 layer to react Li_4SiO_4 , leading to diffusion rate-limited state.

To calculate the CO_2 capture rate, the natural logarithmic scale was used. As shown in **Fig. 5(b)**, there were two slopes at initial stage and final stage, indicating the existence of two rate constants. The rate constant values calculated from the $\ln(C/C_0)$ -time slope are summarized in **Table 1**. The initial (k_1) and final (k_2) rate constants were 0.0075 and 0.0058 (s^{-1}), respectively, which was same value regardless of initial CO_2 concentration. At the initial stage, CO_2 react with Li_4SiO_4 immediately. Thus, k_1 means the reaction rate between CO_2 and Li_4SiO_4 . As a reaction time was longer, the diffusion of CO_2 into Li_2CO_3 layer dominated the reaction. Therefore, the meaning of k_2 is the diffusion rate of CO_2 .

The XRD patterns and SEM images of $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ sample with CO_2 capture at 600 °C and regeneration at 700 °C are shown in **Figs. 6** and **7**, respectively. In the initial $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ sample, Li_4SiO_4 mainly formed beside Li_2SiO_3 and polygonal particles with the size of about 500 nm were observed. With CO_2 capture sample,

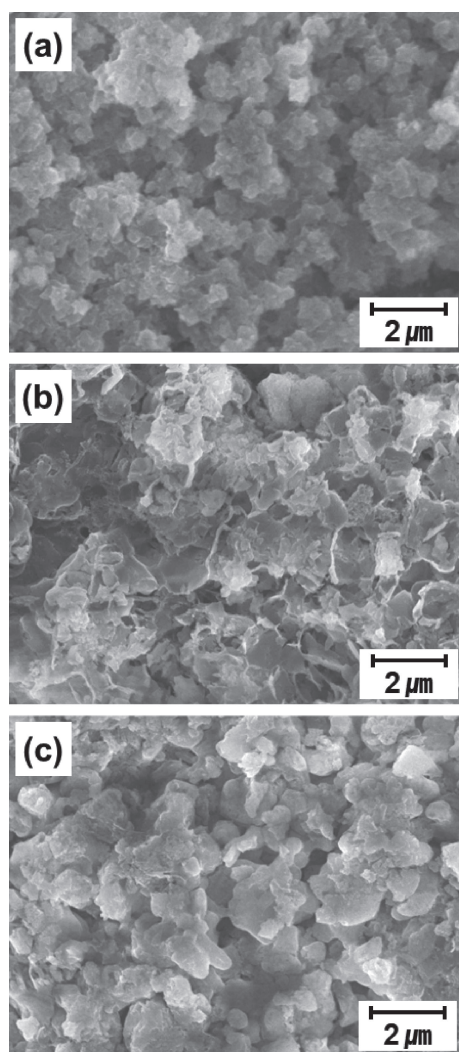


Fig. 7. SEM images of surface of (a) $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ porous sample after (b) CO_2 capture at 600 °C and (c) regeneration at 700 °C.

the peaks of Li_2SiO_3 and Li_2CO_3 were detected and irregular morphology was observed, indicating that CO_2 completely reacts with Li_4SiO_4 . After heat treatment at 700 °C under N_2 flow, Li_4SiO_4 peaks were detected with slight peaks of Li_2SiO_3 , indicating almost all regeneration was achieved. However, particles morphology did not return to initial state, which was with wide size distribution.

Table 2 summarizes the rate constant values obtained in this study and several literatures.^{29)–36)} Compared to other studies, the k_1 in this study was not so much high, which means the reaction rate between CO_2 and Li_4SiO_4 was low. This was due to a large flow rate and low CO_2 concentration in this study. On the other hand, the k_2 was highest among them. This indicates that the diffusion of CO_2 into Li_2SiO_3 and Li_2CO_3 layer was quite accelerated. There are two possible reasons for a high k_2 value. One is a low reaction rate of Li_4SiO_4 and CO_2 . The other is porous structure accelerated CO_2 diffusion. Anyway, we achieved the enhancement of CO_2 diffusion in this study.

3.3 Cycle performance test

The CO_2 capture and regeneration test of $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ sample was measured in several cycles to investigate the cycle performance. **Figure 8** shows the relationship between the number of cycles and the CO_2 capture amount. The test conditions were set with 15 vol % CO_2 at 600 °C for CO_2 capture and with N_2 gas at 700 °C for regeneration. The CO_2 capture ability of $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ per 1 g was 42.2 mg at 1st cycle and kept a stable over several cycles. The CO_2 capture capacity slightly decreased with an increase in cycle number, reaching 40.9 mg at the 20th cycle. This decrease in CO_2 capture ability was due to the particle growth during heat treatment at high temperature like Fig. 7.

4. Conclusions

SiO_2 porous glass modified with Li_4SiO_4 can be successfully development by the solid-state reaction method

Table 2. Kinetic constants compared to literatures

Sample feature	Temperature (°C)		Flow rate (ml/min)		Composition (vol %)		k_1 (s^{-1})	k_2 (s^{-1})	Ref.
	CO_2 sorption	CO_2 desorption	CO_2 sorption	CO_2 desorption	CO_2 sorption	CO_2 desorption			
Li_2SiO_3 , $\text{Li}_3\text{NaSiO}_4$ powder	575	>650	—	—	CO_2 : 100	N_2 : 100	0.0050	0.0002	29)
Li_2SiO_3 , Li_4SiO_4 powder	650	750	50	50	CO_2 : 100	N_2 : 100	0.0420	0.0010	30)
Li_4SiO_4 powder	550	550	50	50	CO_2 : 100	N_2 : 100	0.0088	0.0002	31)
Li_4SiO_4 powder	550	650	60	60	CO_2 : 15	N_2 : 100	0.0650	0.0061	32)
LiOH , Li_2SiO_3 , Li_4SiO_4 powder	700	725	100	100	CO_2 : 60	N_2 : 100	0.0220	0.0004	33)
Li_2SiO_3 , Li_4SiO_4 powder	610	700	50	50	CO_2 : 30	Ar: 100	0.0230	0.0011	34)
Li_2SiO_3 , Li_4SiO_4 powder	590	700	60	60	CO_2 : 50	N_2 : 100	0.0140	0.0006	35)
Li_2CO_3 , Li_2SiO_3 , Li_4SiO_4 powder	650	>700	—	—	CO_2 : 60	N_2 : 100	0.0654	0.0016	36)
$\text{Li}_4\text{SiO}_4/\text{SiO}_2$ porous body	600	700	200	200	CO_2 : 0.2–10	N_2 : 100	0.0075	0.0058	This work

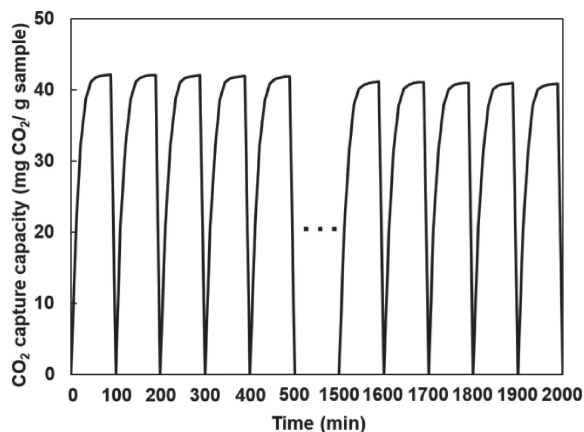


Fig. 8. CO_2 capture cycle performance of the $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ sample at 100 mL/min of flow rate. The test conditions were set with 15 vol% CO_2 at 600 °C for CO_2 capture and with N_2 gas at 700 °C for regeneration.

with CO_2 capture ability at high temperature. The $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ capture material was confirmed to have a low reaction rate but quite high diffusion rate. The CO_2 capture ability was stable over several cycles with a capture capacity of 42.2 to 40.9 mg per gram for 20 cycles at 600 °C. Li_4SiO_4 completely reacted with CO_2 and was completely regenerated into the original Li_4SiO_4 phase at 700 °C for several cycles. The porous $\text{Li}_4\text{SiO}_4/\text{SiO}_2$ capture material developed in this study has a high potential in applications for high-temperature CO_2 capture.

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