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A Statistical Analysis of Sputtering Parameters on Superconducting Properties of Niobium Thin Film

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Abstract: With everchanging technology and constantly increasing demand of energy, superconducting materials can play a crucial role as by offering lowest possible electrical resistance. In this quest, Niobium (Nb) thin films synthesized by magnetron sputtering has been analyzed at diverse sputtering constraints. Different parameters as Nitrogen (N₂) partial pressure, power densities, magnetization are explored by Usadel equation. Usadel equation integrating Bardeen-Cooper-Schrieffer theory of superconductivity, **analyzed dependence of superconducting transition** temperature on varied sputtering parameters by computational technique. It has been found that Niobium Nitride (NbN) operating at elevated temperature exhibits superconductive properties. Optimal growth is found at N₂ partial pressure ~15 Standard Cubic Centimeters per Minute (SCCM) where power densities ranges from 200 to 500W at a critical temperature of 14.5K.

Keywords: Sputtering; Superconducting; Thin film; Growth

1. Introduction

For state of art scientific research, potential of new superconducting materials are explored which can operate at elevated temperatures and presents new industrial applications where Niobium is found extremely suitable¹⁻³. Thin films grown of superconductor materials on suitable substrate presents possibility of enormous range of devices exhibiting superconducting properties at varied operational ranges viz. temperature, magnetic field, voltage etc.⁴⁻⁶. In today's world of energy deprived world, it is of great interest to identify or develop engineered materials capable of high performance applications at reduced energy consumption⁷. Thin film growth, often multilayer⁴, of superconducting material presents a promising technique of obtaining high performance superconducting cavities. Such coated superconductors in the form of superconducting nanowire single photon detectors (SNSPDs), superconducting radiofrequency (SRF) cavities, Superconductor-Insulator-Superconductor (SIS) tunnel junctions⁸⁻¹² presents a huge potential for diverse applications ranging from electronic industry to aerospace industry and industrial applications. As reduced power consumption like in power cables, magnets, motors, quantum information, quantum optics, free space laser communication, quantum electronics, particle accelerators, etc.^{3,5,13}.

For the above varied applications Transition Metal

Nitrides (TMN) has been widely accepted. Many studies were focused on Niobium, Titanium, Lead, Bismuth & Tin as superconductive coating^{9,14-17}. For depositing thin films various techniques of sputtering are employed such as pulsed DC, pulsed laser deposition, RF and DC sputtering¹⁸⁻²⁰ chemical vapor deposition (CVD)¹¹ Ion Beam Sputtering²¹ are used to obtain stable thin film of superconductors, also these techniques are deployed to fabricate Spintronic devices. The spintronic devices has superior performance wherein same semiconductor charge, spin of moving electron is manipulated.

Deposited superconductors on a substrate involves following steps (a) substrate preparation- involving superfinishing operations, (b) buffer / isolating layer, (c) superconducting layer (sputtering technique), (d) protective layer.²²⁻²⁴ Stability of such films poses great challenge alongwith performance specially under the effect of temperatures, magnetic field, electrical resistivity etc. Researchers have been studying characteristics, homogeneity etc. of film deposited on various substrates (MgO, Silicon, sapphire) which shows a great dependence on technology of deposition and parameters chosen for deposition^{25,26}.

Matsumoto *et al.*,⁵ fabricated thin film Bi-Sr-Ca-Cu-O (BSCCO) and successfully doped Bi, Pb-2223 in stable phase using RF sputtering to investigate superconducting properties. Where an significant

improvement of around 10.8 K in transition temperature (T_c) were observed. Xiaoyan *et al.*,¹⁵⁾ obtained thin film of Niobium Titanium Nitride (NbTiN) and NbN on Si wafers and concluded that NbTiN stands out in performance (low resistivity at elevated temp., low kinetic inductance, reduced recovery time of photon response, higher switching currents (I_{sw}), low timing jitter (TJ), kinetic inductance) and uniformity as compared to NbN. Tomas polakovic *et al.*,⁸⁾ shows : Growth of film depends on concentration of gases and sputtering powers which in turn needs precise control. In DC magnetron sputtering employing Nb target in presence of neutralized nitrogen is capable of producing NbN thin films with higher T_c (superconducting transition temperatures) 14.5K and resistivity of 110.62 $\mu\Omega\cdot\text{cm}$.

A. Shoji *et al.*²⁷⁾ finds that Reactive sputtering is capable of producing desired stable phases of NbN at temperatures more than 773K whereas higher T_c of NbN film can be obtained by using methane gas in RF diode sputtering at 300K²⁸⁾ and gives high resistivity above $10^4 \mu\Omega\cdot\text{cm}$ due to presence of columnar or granular structure²⁹⁾. Also Bombarding ion at energies above 300 eV can cause damage to structure of thin film as claimed by W. Kern *et al.*³⁰⁾ also they showed that their exits superconducting T_c at 12.5 K or > 12 K for concentration of nitrogen ranging from 7.5% to 13% .

The superconducting state of NbN suppresses as the deposited film thickness reaches to thin film critical thickness, for superconducting thin films critical thickness is 2.7 nm below which NbN thin film are not superconducting^{31,32)}. However, influence of sputtering parameters, primarily on thickness and other properties is still an open question and researchers are worldwide experimenting towards it. As cited above Deposition parameters and growth method has significant effect on structural properties.

Niobium Nitride presents a promising element acting as thin layer of superconducting element. Which is comparatively easy to fabricate along with large superconducting energy gap, relatively high superconducting (T_c) temperature thus showing superconducting properties at elevated temperatures.

Fig. 1: Schematic picture of sputtering chamber

A magnetron sputtering device as shown in Fig. 1 is considered wherein Nitrogen plasma parameters on

relative thickness of film over a substrate surface is computed.

Here in this study superconductive nature of Niobium thin film and their synthesis by Magnetron sputter deposition upon substrates at different parameters as Nitrogen (N_2) partial pressure, power densities, magnetization are investigated by computational technique using Usadel equation for a superconductive state.

2. Materials and methods

Magnetron sputtering method employs bombardment of target element by ions at high energy and velocity under the influence of electric and magnetic field in the presence of nitrogen plasma. Figure 1 shows device for sputtering which is considered for deposition of Nb as thin film with use of Nitrogen gas as plasma, ions extracted from plasma bombards target surface, as positioned very near to plasma, this bombardment results in dislodging of target atoms and is directed towards substrate and makes a thin layer over bulk material. Plasma ions spot size 2-6mm and diameter of column is taken as 4 cm with ion velocity derived from real frequency of waves^{33,34)} which further depends on electromagnetic lenses focusing. Sputtering chamber pressure is considered 2×10^{-5} Pa^{8,15)}, flow of gas 10-20 Standard Cubic Centimeters per Minute (SCCM) , energies above 100 ev. Ratio of threshold energy by ions energy are adopted from Grais *et. al.*³⁵⁾. Relative thickness of film is calculated by following relation³⁶⁾

$$T_p = h^2 (T_{p1} - \alpha h^2 T_{p2}) \quad (1)$$

$$T_c = T_{c1} - \alpha T_{c2} \quad (2)$$

Where T_{p1} , T_{p2} , T_{c1} , T_{c2} is as follows: -

$$T_{p1} = 2 \int_{R_1}^{R_2} R dR \int_0^\pi \frac{d\phi}{L_0^5 (1 - K \cos \phi)^{5/2}} \quad (3)$$

$$T_{p2} = 2 \int_{R_1}^{R_2} R dR \int_0^\pi \frac{d\phi}{L_0^7 (1 - K \cos \phi)^{7/2}} \quad (4)$$

$$T_{c1} = \int_h^{h+1} h dh \int_0^{2\pi} R_2 d\phi \frac{(R_2 - r \cos \phi)}{L_0^5 (1 - K \cos \phi)^{5/2}} \quad (5)$$

$$T_{c2} = \int_h^{h+1} h dh \int_0^{2\pi} R_2 d\phi \frac{(R_2 - r \cos \phi)^3}{L_0^7 (1 - K \cos \phi)^{7/2}} \quad (6)$$

Real frequency of waves can be obtained given by Tyagi *et.al.*,³⁴⁾

$$\text{With } K = \frac{2Rr}{L_0^2} \text{ and } L_0^2 = R^2 + r^2 + h^2$$

Niobium (Nb) offers lower surface resistance R_s at high lower critical field B_c in the presence of large energy gap Δ , however coherence length ξ become shorter for higher critical field B_c as shown by eq. 7 & 8

$$R_s = \frac{A\omega^2}{T} \exp \left(-\frac{\Delta}{T} \right) \quad (7)$$

Neglecting residual stresses

$$B_c = \frac{2\Phi}{\pi d^2} \ln \frac{d}{1.07 \xi} \quad D < \lambda \quad (8)$$

Calculating transition temperature T_c affected by different parameters to a considerable value saves a lot of time and effort wasted by trial and error method. These estimated values provide a reliable control so as to accommodate experimental results. Particularly work of Martinis et al.³⁷⁾ provide a simplified result of Usadel Equation³⁸⁾ which in turn is based on Bardeen-Cooper-Schrieffer (BCS) theory of superconductivity, valid in the diffusive limit. Usadel equation for a superconductive state represented by function $\theta(x, E)$ can be idealized as

$$iE \sin \theta + \Delta(x) \cos \theta = 0 \quad (9)$$

here this is approximated for uniform superconductor x in one dimension only. θ in the range of 0 to $\pi/2$.

Where,
 $\cos \theta$ is replaced by $\cos \theta_{BCS} = |E|/\sqrt{E^2 - \Delta^2}$
and $\sin \theta$ by $\sin \theta_{BCS} = \Delta/\sqrt{\Delta^2 - E^2}$

In superconducting region Δ is taken as constant. Taking clue from BCS gap equation

$$T_c = T_{co} \left[\frac{\alpha}{(1.13 \alpha + 1.13) t} \right]^\alpha \quad (10)$$

Where $\alpha = d_n n_n / d_s n_s$, d_n and d_s are thickness of normal and superconducting film, n_s and n_n are density of electronic state in normal and superconducting film.

3. Results and discussion

To understand and demonstrate relative dependence of film thickness on Nitrogen concentration and to facilitate comparison of superconducting transition temperature and its range exhibiting superconductivity is computed and analysed with the help of mathematical model and computer technique by means of selectively chosen plasma parameters as shown in Table 1. Results are found in good proximity with literature.

Table 1. Coating procedure parameters - current 1-1.20 A, with deposition time 6- 60 per minute.

T_c (K)	N (at. %)	N ₂ P%	Nb
16.55	42.5	18	41.7
16.67	46.1	21	37.7

Fig. 2: Thickness versus SCCM nitrogen pressure

Fig. 2 shows thickness of NbN film at varied power densities under the influence of nitrogen pressure. Film thickness follows a steady increase and decrease after reaching an optimal thickness. After a critical range it presents insensitivity towards nitrogen in the deposition chamber.

As inferred by literature though plasma condition directly affect deposition of target material on a substrate and is also evident from the figure where at higher power substantially higher deposition rates are obtained. Maximum growth is achieved in a range of 13 to 19 SCCM, which observed approximately 18% growth in small window and almost getting stagnant and insensitive to further increase in nitrogen.^{27-29,39)} with increased power at a single point of observation that is at some fixed N₂ partial pressures, thickness or growth rate enhances due to increase of sputtering rate (Nb), further effecting on stability of NbN Phases due to increased Nb/N ratio. This optimum window provide maximum variation in thickness of film as due to epitaxial strain superconducting properties get affected for very thin film.

Fig. 3: Temp (K) versus magnetization showing superconductivity

The influence of temperature on superconductivity is given in Fig. 3 Dashed lines are guiding line representing average mean of corresponding values. The computed results shows that there exists superconductivity at temperature ranging from 3K to 14K. Superconductivity is analyzed and shown by the values as measure of magnetic moment at low temperature at subsequent range. Practically it is done by placing samples in constant magnetic field and then sample is demagnetized, and values obtained of magnetic moment ~~and~~ are analyzed. Our results confirm a close relationship between the shape of magnetic hysteresis loop obtained by Jirsa et. al and Mikitik et. al ^{40,41}. The deviation from experimental Magnetic Hysteresis loop (MHL) might be due to presence of phases and impurities at high fields. Increase in critical currents and activating non superconducting particles due to additional pinning and due to interplay of different pinning regimes might lead to fishtail phenomena ⁴²⁻⁴⁵. Free electron density could be a source of error in computed results. Different power intensity has not such a significance effect on superconducting properties, but a critical thickness exists that is must for presenting compositional superconductors. Figure 3 shows response close to diamagnetic response at temperature $\sim 8\text{K}$ to $\sim 11.5\text{K}$ as experimentally proved by Zou et al. ⁴⁶ though not confirming exactly to their findings our reason for that is due to presence of $\delta\text{-NbN}$ in hexagonal and cubic form where cubic $\delta\text{-NbN}$ is mechanically unstable and criteria is not included in calculations.

Fig. 4: Critical temperature (T_c) versus thickness (nm)

Fig. 4 shows variation of critical temperature T_c at constant power but varying thickness of film, Dashed lines are guiding line representing average mean of corresponding values. considering conditions near stoichiometry from 2.3K to 6.5K. as film growth showing insensitivity after certain pressures while taking case of maximum and minimum growth at a power density. The maximum T_c obtained is less than maximum critical

temperature ⁴⁷) at a perfect stoichiometric ratio though confirming the pattern and critical field at $\sim 4.4\text{K}$. At fermi level T_c shows sensitivity due to phases and density of states and might be the reason for the change. The dependence of thickness over T_c for films were analysed by considering the electrical resistance as function of temperature, coherence length value for film is taken from electron mean-free path. For a narrow window range of transition temperature reversal of values is observed as shown confirming in good agreement to Destraz et.al., ⁴⁸)

It seems that instead of disordered scattering T_c , in different growth conditions and film, is governed by change in stoichiometric compound. While decrease in sputtering power there is an indication of decrease in T_c as with reducing power Nb/N ratio is affected and carrier density as function of stoichiometric compound decreases.

Superconducting T_c goes down as the local density is suppressed by grain boundary ^{49,50}) at different levels of T_c , by analyzing carrier density based on hall effect in thin film can be explored to resolve above issue not done here will be stated elsewhere. Transition temperature shows a similar behavior for a range of film thickness.

4. Conclusion

Here in this study thin films of Nb were deposited upon substrates at different N_2 partial pressures and NbN films are investigated by computational technique using Usadel equation for a superconductive state. Film developed at 200W and 500W covers maximum range of nitrogen pressure and presents region for maximum film growth variation. Film at these power exhibits superconductivity in the range of 6.2 K to 13 K for the range of 0.078 emu to 0.02 emu, while having most stability at higher power 500W. At higher fields due the presence of impurities and pinning effect an deviation from computed Magnetic Hysteresis Loop is observed during experimental findings as probably by activation of non-superconducting particles and due to the influence of free electron density. Superconducting properties are not significantly affected by power density. Nb thickness reduced with decrease in superconductivity transition temperature. Film growth after certain pressures does not shows much dependence on the maximum critical temperature T_c whereas T_c maximum representing maximum growth is well below critical temperature at perfect stoichiometric ratio. This phenomenon is due to combined effect of film resistance, film structure and growth at particular thickness.

Thin NbN film grown over a substrate is probed by Usadel equation for close fitting to experimental data available. Usadel equation incorporating BCS theory of superconductivity relates decrease in film thickness to increase in BCS coherence length $\xi \sim 3.2\text{ nm}$ at a temperature approaching to critical value 14.5 K and magnetization value of 0.05 emu. Affecting stability of NbN Phases with increase of sputtering rate (Nb) corresponding to respective Nb/N ratio.

Nomenclature

T_p	-	Relative thickness of the film for planar magnetron sputtering
T_c	-	Relative thickness of the film for cylindrical magnetron sputtering
α	-	Thermal velocity of ions
R	-	Radius of plasma column
r	-	Radius of plasma nozzle
H	-	Substrate to target height
Φ	-	Azimuthal angle
R_s	-	Surface resistance
B_c	-	Critical field
ξ	-	Coherence length

$\omega = \omega_r + i\gamma$ - Where ω_r =real frequency, γ =growth rate.

T_c	-	Superconducting Transition temperature
$E(x)$	-	Magnitude of homogeneous Direct Current electric field w.r.t distance x from origin
E_{0x}	-	Magnitude of homogeneous Direct Current electric field at origin
e_s	-	Charge of species
n, p	-	Order of Bessel function
x	-	Distance from origin
Ω_i	-	Gyro frequency
$Z(\xi)$	-	Plasma dispersion function

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