

## Assessment of hydrogen embrittlement via image-based techniques in Al-Zn-Mg-Cu aluminum alloys

Su, Hang

Department of Mechanical Engineering, Kyushu University

Toda, Hiroyuki

Department of Mechanical Engineering, Kyushu University

Shimizu, Kazuyuki

Department of Mechanical Engineering, Kyushu University

Uesugi, kentaro

Japan Synchrotron Radiation Research Institute(JASRI)

他

<https://hdl.handle.net/2324/4149932>

---

出版情報 : Acta Materialia. 176 (1), pp.96-108, 2019-09-01. Elsevier  
バージョン :  
権利関係 :



## **Figures and tables**

Table.1 Chemical compositions of the three alloys (mass %)

|            | Si   | Fe   | Cu   | Mg   | Zn    | Ti   | Zr   | Al  |
|------------|------|------|------|------|-------|------|------|-----|
| High Fe Si | 0.30 | 0.30 | 1.50 | 2.40 | 10.00 | 0.04 | 0.15 | Bal |
| Low Fe Si  | 0.01 | 0.01 | 1.50 | 2.40 | 10.00 | 0.04 | 0.15 | Bal |

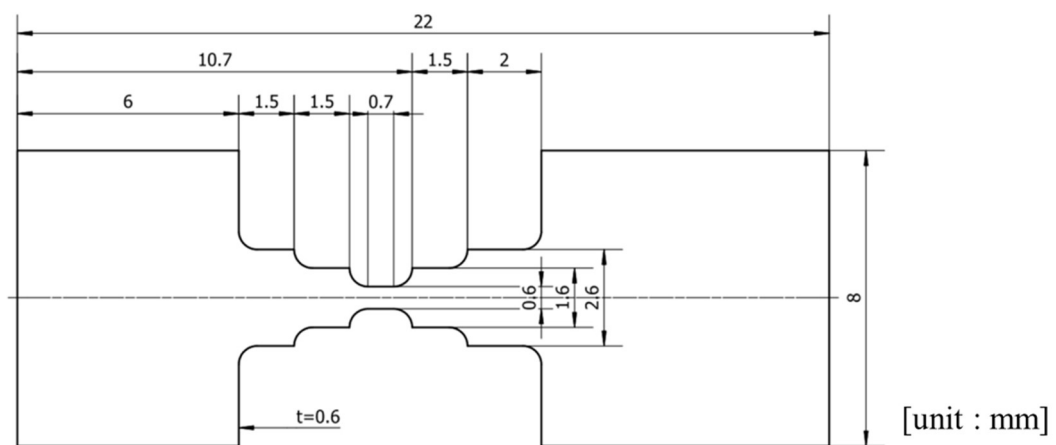


Fig.1 Geometry of an in-situ tensile test specimen

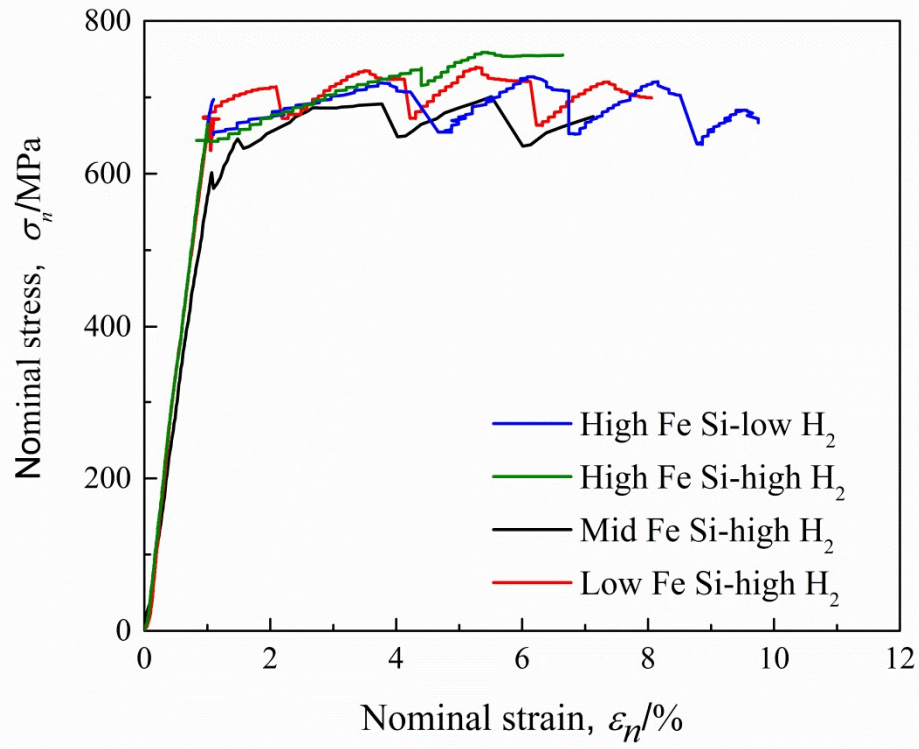


Fig. 2 Nominal stress-strain curves for the in-situ tensile tests. It is worth noting that nominal stress-strain curve of Mid Fe Si-high  $H_2$  has already been applied in Ref [15].

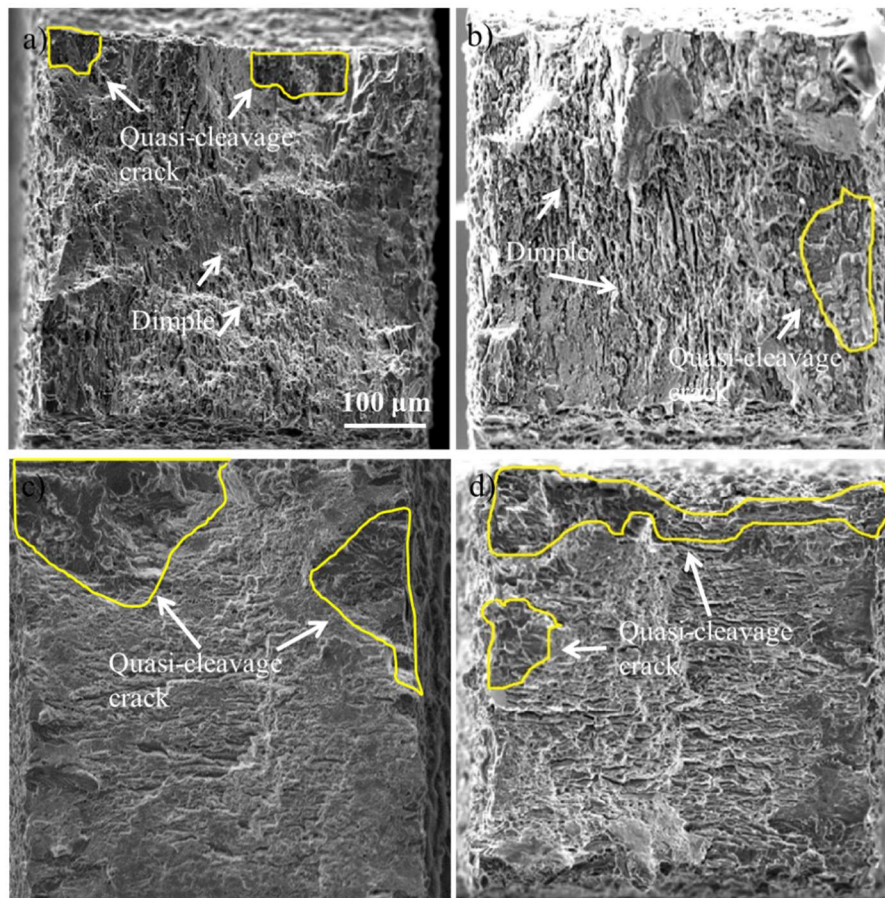


Fig.3 Fracture surfaces after the in-situ tensile tests; a) High Fe Si-low  $H_2$  specimen, b) High Fe Si-high  $H_2$  specimen, c) Mid Fe Si-high  $H_2$  specimen [15] and d) Low Fe Si-high  $H_2$  specimen; the quasi-cleavage crack region is identified by yellow solid lines.

Table.2 Areal fractions of the quasi-cleavage cracks

| Materials                      | Areal fraction [%] |
|--------------------------------|--------------------|
| High Fe Si-low H <sub>2</sub>  | 5.7                |
| High Fe Si-high H <sub>2</sub> | 8.1                |
| Mid Fe Si-high H <sub>2</sub>  | 18.8 [15]          |
| Low Fe Si-high H <sub>2</sub>  | 22.4               |

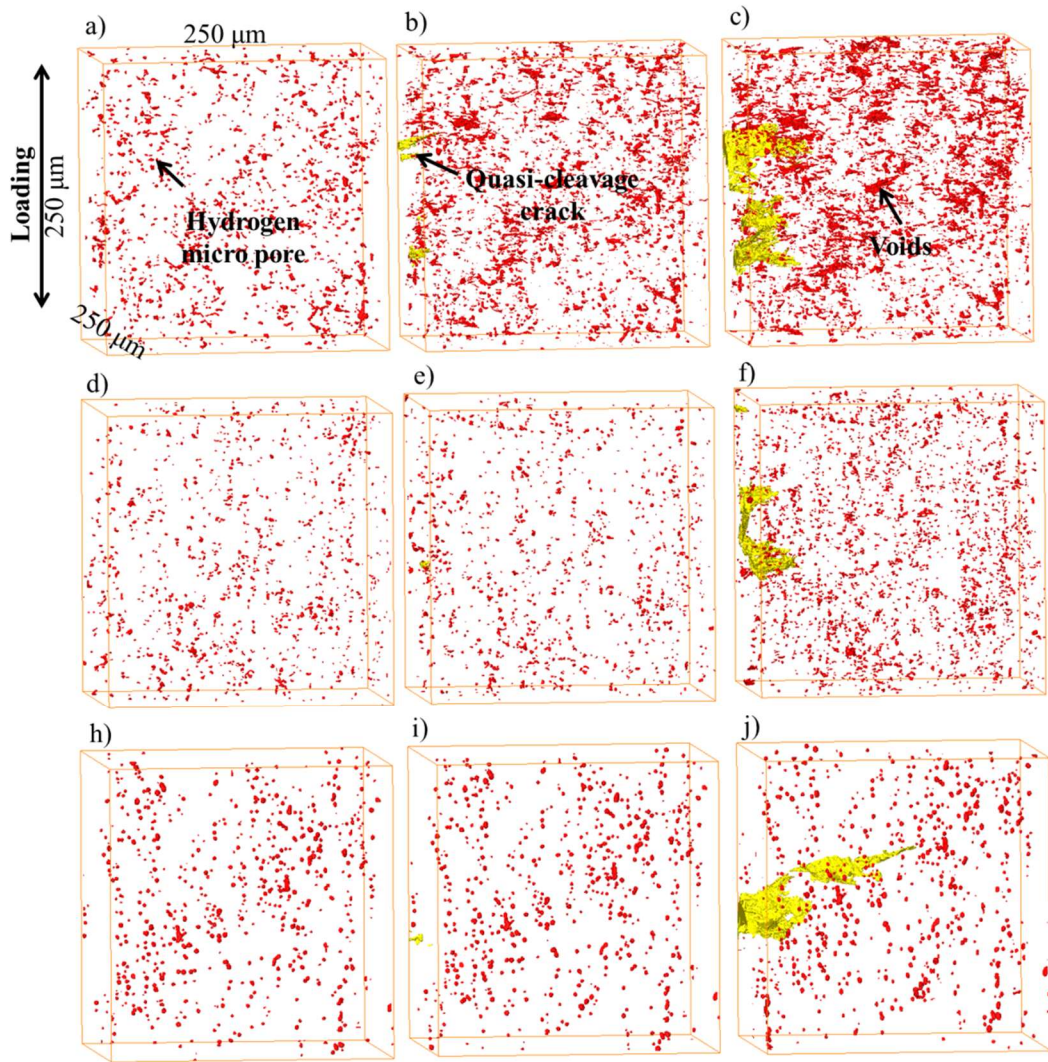


Fig.4 4D observations of the initiation and propagation of the quasi-cleavage crack in High Fe Si-low H<sub>2</sub> under different applied strains; a)  $\varepsilon_a = 0.0 \%$ , b)  $\varepsilon_a = 6.3 \%$  and c)  $\varepsilon_a = 8.6 \%$ ; in High Fe Si-high H<sub>2</sub> under different applied strains; d)  $\varepsilon_a = 0.0 \%$ , e)  $\varepsilon_a = 1.0 \%$  and f)  $\varepsilon_a = 4.4 \%$  and that in Low Fe Si-high H<sub>2</sub> under different applied strains; h)  $\varepsilon_a = 0.0 \%$ , i)  $\varepsilon_a = 2.4 \%$  and j)  $\varepsilon_a = 6.9 \%$  Hydrogen micro pores are shown in red and the quasi-cleavage crack is shown in yellow



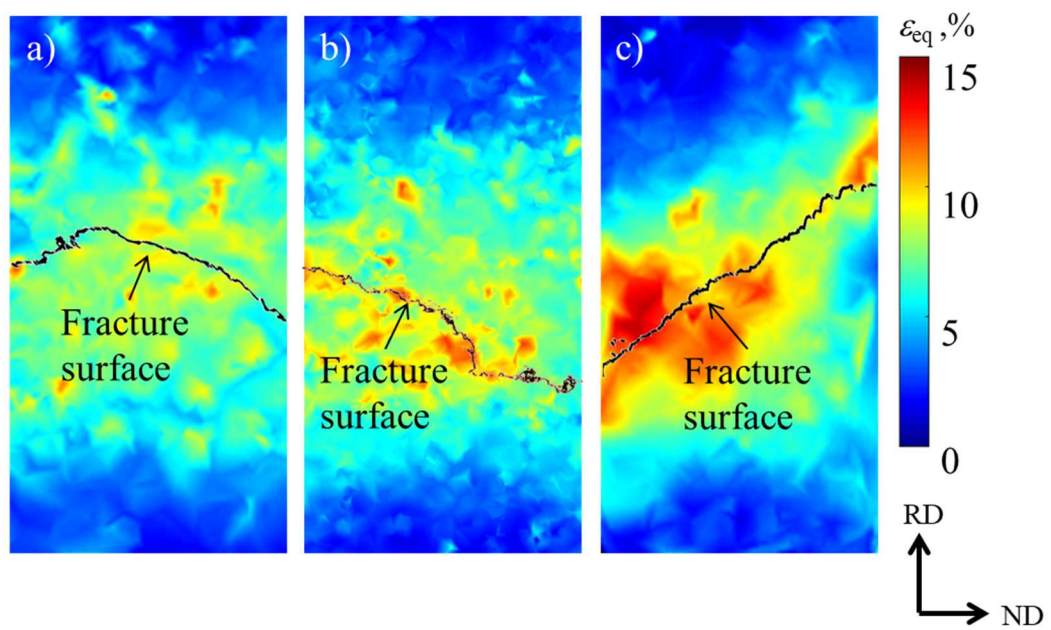


Fig. 5 The equivalent strain ( $\epsilon_{eq}$ ) distribution under different applied strains, viewed on the y-z (RD-ND) cross-section; a) equivalent strain map calculated between  $\epsilon_a$  of 2.1 and 6.3 % in High Fe Si-low  $H_2$ , b) equivalent strain map calculated between  $\epsilon_a$  of 0 and 4.4 % in High Fe Si-high  $H_2$  and c) equivalent strain map calculated between  $\epsilon_a$  of 2.4 and 6.9 % in Low Fe Si-high  $H_2$ . Fracture surface is shown as the black line.

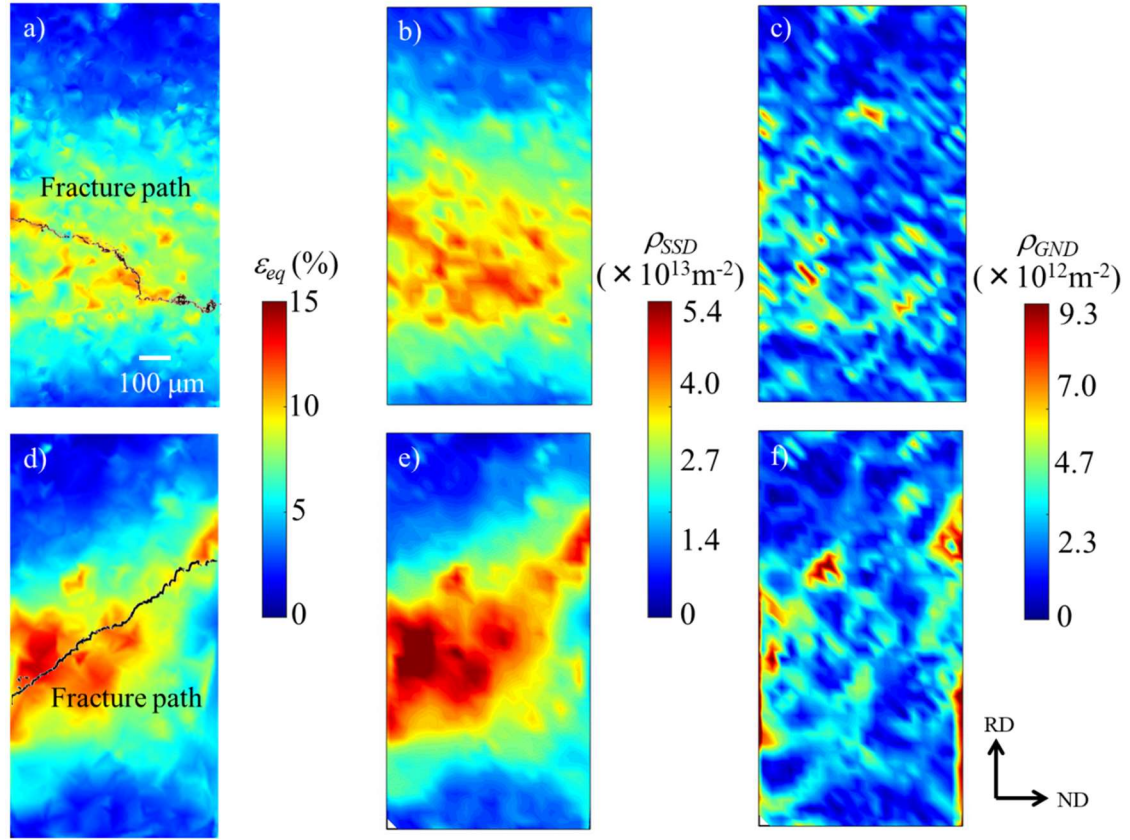


Fig. 6 SSD and GND densities calculated from the equivalent strain ( $\epsilon_{eq}$ ) mapping under different applied strains, viewed on the y-z (RD-ND) cross-section; a) equivalent strain map calculated between  $\epsilon_a$  of 2.1 and 6.3 % in High Fe Si-high  $H_2$ , b) SSD density calculated from a), c) GND density calculated from a); d) equivalent strain map calculated between  $\epsilon_a$  of 2.4 and 6.9 % in Low Fe Si-high  $H_2$ , e) SSD density calculated from d) and f) GND density calculated from d). Fracture path is shown as the black line in a) and d).

Table. 3 Binding energy of each hydrogen trap site in Al-Zn-Mg-Cu aluminum alloys

| Trap Site         | Binding energy<br>$E_b/\text{KJmol}^{-1}$ | Trap Site               | Binding energy<br>$E_b/\text{KJmol}^{-1}$ |
|-------------------|---|-------------------------|---|
| Edge dislocation  | 9.64[25]                                  | Precipitates            | 33.77[26]                                 |
| Screw dislocation | 7.72[25]                                  | Intermetallic particles | 28.10[43]                                 |
| Vacancy           | 28.95[24]                                 | Pore                    | 67.20[44]                                 |
| Grain boundary    | 19.30[24]                                 |                         |   |

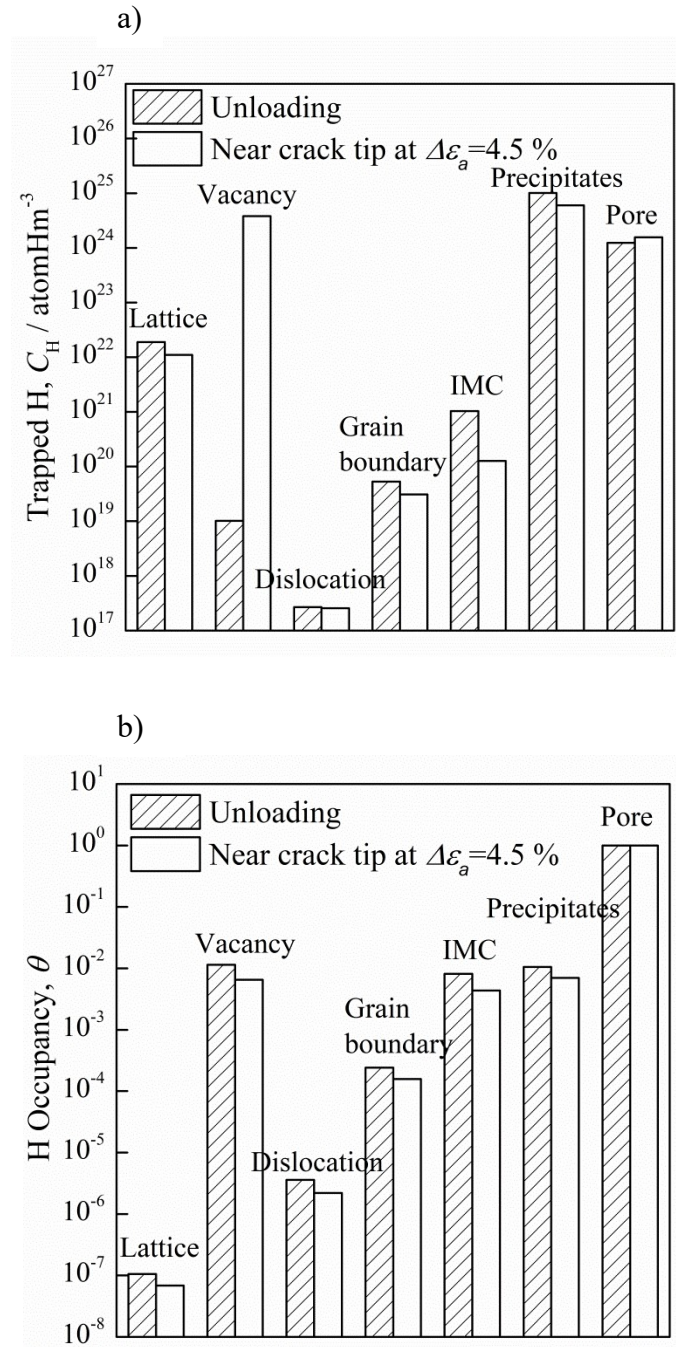


Fig. 7 Comparisons of the hydrogen trapping behavior between the whole specimen at unloading state and a region ahead of the hydrogen-induced quasi-cleavage crack tip an applied strain of 4.5 % in Low Fe Si-high H<sub>2</sub> specimen; a) trapped hydrogen content and b) hydrogen trapping occupancy

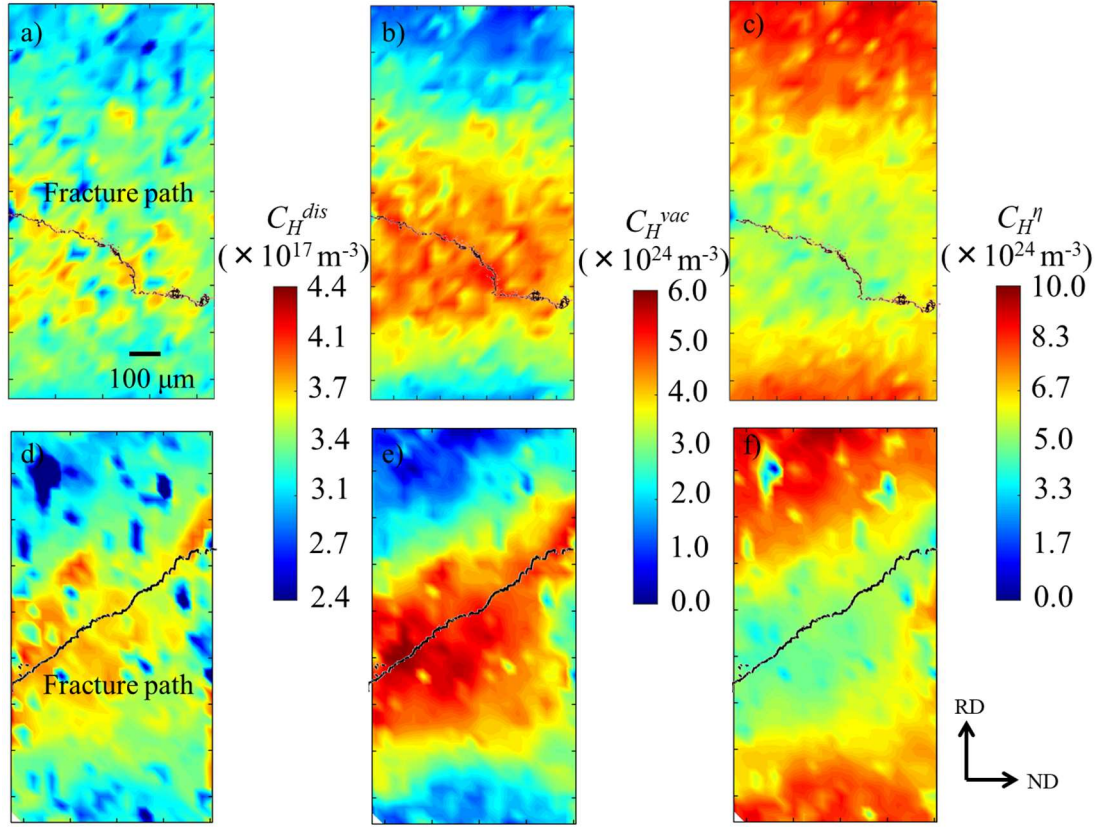
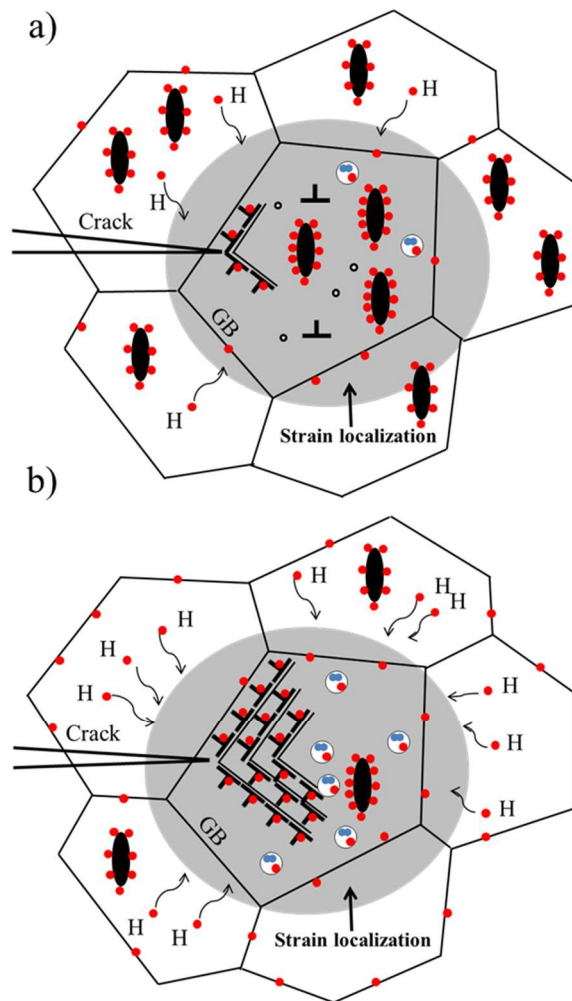


Fig. 8 Hydrogen concentrations at various trap sites including dislocations, vacancies and precipitates that calculated from the equivalent strain ( $\epsilon_{eq}$ ) mapping at an applied strain,  $\Delta\epsilon_a$  of approximately 4.5 %, viewed on the y-z (RD-ND) cross-section; a) and d) is the hydrogen concentration at dislocations in High Fe Si-high H<sub>2</sub> and Low Fe Si-high H<sub>2</sub> specimen, b) and e) is the hydrogen concentration at vacancies in High Fe Si-high H<sub>2</sub> and Low Fe Si-high H<sub>2</sub> specimen and c) and f) is the hydrogen concentration at precipitates in High Fe Si-high H<sub>2</sub> and Low Fe Si-high H<sub>2</sub> specimen.

Fracture surface is shown as the black line.





• Vacancy    Particle/precipitate    Dislocation    Nano void    • H    • H<sub>2</sub>

Fig.9 Schematic illustration of the influence of intermetallic particles to the hydrogen partitioning and related crack propagation in the strain localization region, a) high intermetallic particle content material and b) low intermetallic particle content material